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# Magnetic and electrical properties of  $Fe_{2+x}V_{1-x}Al$

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# **Abstract**

Magnetization, electrical resistivity, thermoelectric power and magnetoresistance were measured for the ordered alloys  $Fe_{2+x}V_{1-x}Al$  $(0 \le x \le 0.4)$ . The magnetization measurement shows that the phases are ferromagnetic. Both the Curie temperature and the spontaneous magnetization increase with increasing *x*. The samples with  $x \le 0.1$  exhibit a semiconductor-like behavior. Energy gap  $E_G$  decreases with increasing *x*. The electrical resistivity versus temperature curves make a broad maximum near  $T_c$  for the samples with  $x=0.18$ , 0.20 and 0.40. For Fe<sub>2.18</sub>V<sub>0.82</sub>Al, negative large magnetoresistance is observed.  $\circ$  2001 Elsevier Science B.V. All rights reserved.

*Keywords:* Fe<sub>2+x</sub>V<sub>1-x</sub>Al; Spontaneous magnetization; Curie temperature; Electrical resistivity; Magnetoresistance; Thermoelectric power

## **1. Introduction**

Webster and Ziebeck reported that  $Fe<sub>2</sub>VAL$  forms a single-phase intermetallic compound with the Heusler  $L2<sub>1</sub>$ structure [1]. The crystal structure of  $Fe<sub>2</sub>$ VAl is shown in Fig. 1. According to their experimental results of the magnetic measurement,  $Fe<sub>2</sub>$ VAl remains weakly paramagnetic in the temperature range from 5 to 300 K. This paramagnetic property of  $Fe<sub>2</sub>VAL$  was shown to be non-Curie–Weiss. Recently,  $Fe<sub>2</sub>$ VAl has attracted the interest because of the intriguing behavior of its transport, photoelectric and thermodynamic properties [2–7]. The resistivity shows semiconducting behavior with a negative temperature coefficient suggesting an energy gap of  $\sim 0.1$  eV [2]. The photoemission spectrum, on the other hand, shows a clear Fermi edge characterizing the metallic properties of Fe, VAI [2]. Low-temperature specific heat measurements revealed an unusual upturn in *C*/*T* with decreasing temperature, commonly observed in most heavy-fermion systems [2,7]. From band structure calculations,  $Fe<sub>2</sub>VAL$  is a semimetal with a pseudogap at the Fermi level [8–11].

Intermetallic compound  $Fe<sub>3</sub>Al$  is a ferromagnet with a  $DO<sub>3</sub>$  crystal structure. The Curie temperature  $T<sub>C</sub>$  and the saturation magnetic moment at 4.2 K of Fe<sub>3</sub>Al are  $\sim$ 770 K [12] and  $\sim$  5.2  $\mu_B$ /f.u. [13], respectively. The mixed crystals  $Fe_{2+x}V_{1-x}Al$  ( $0 \le x \le 1$ ) of  $Fe_2VAl$  and  $Fe_3Al$  form Fig. 1. Unit cell of the cubic Heusler-type compound Fe, VAI. The sites



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are represented by A, B, C and D. The A and C sites are equivalent and \*Corresponding author. Fax: 181-22-368-7070. are occupied by Fe atoms, the B site is occupied by V, and the D site by

a complete solid solution [14]. Nishino et al. [2] and Endo et al. [3] found that  $Fe_{2+x}V_{1-x}Al$  with  $1 \ge x > 0$  shows ferromagnetism, and a curious 'resistance maximum' near  $T_c$  in the electrical resistivity  $\rho$  versus temperature curves. At temperatures above  $T_c$ , the temperature gradient of  $\rho$ was negative, similar to that of semiconductor. Moreover, Endo et al. found that the absolute value of the transverse magnetoresistance represented by  $\Delta \rho / \rho(0)$  takes maximum near  $T_c$  for Fe<sub>2.2</sub>V<sub>0.8</sub>Al, where  $\Delta \rho$  is  $\rho(H) - \rho(0)$ , and  $\rho(0)$  is zero field resistivity at the measured temperature. The value of  $\Delta \rho / \rho(0)$  for Fe<sub>2.2</sub>V<sub>0.8</sub> Al is negative and ~5% at 13 kOe near  $T_{\rm C}$ . Such resistance phenomena have not yet been explained. To make clear the origin of these transport properties, we have studied systematically the magnetic and electrical properties of Heusler-type pseudobinary alloys  $Fe_{2+x}V_{1-x}Al$  ( $0 \le x \le 0.4$ ).

# **2. Experimental**

The pseudobinary alloys  $Fe_{2+x}V_{1-x}Al$  ( $0 \le x \le 0.4$ ) were<br>prepared by repeated melting of appropriately composed<br>prepared by repeated melting of appropriately composed mixtures of 99.95% pure Fe, 99.9% pure V and 99.9999% pure Al, in an argon arc furnace. Since the weight loss electric contact. The thermoelectric power *S* was measured after melting was negligible, the nominal composition was by the same method as that of Nagy and Toth [18]. The accepted as being accurate. To get the homogenized transverse magnetoresistance measurement, using the stansamples, the reaction products were sealed in evacuated dard four-probe DC technique, was performed in the silica tubes, heated at  $1000^{\circ}C$  for 3 days and then magnetic fields up to 220 kOe using a pulse magnet. quenched in water. X-ray diffraction spectra were taken with  $Cu K<sub>\alpha</sub>$  radiation on powder samples prepared as above. All X-ray lines of the prepared samples were **3. Results and discussions** indexed with the cubic structure. The lattice parameters were determined from the X-ray diffraction patterns using Fig. 3 shows the magnetization curves at 5 K of the Nelson and Rily method [15]. The lattice parameter Fe<sub>2+x</sub>V<sub>1-x</sub>Al ( $x=0.05$ , 0.18, 0.25 and 0.40) in magnetic was found to be  $a=5.7629$  Å for Fe<sub>2</sub>VAl ( $x=0$ ). The fields up to 50 kOe. As seen in Fig. 3, the magneti value of a for Fe<sub>2</sub>VAI is consistent with those previously of samples is saturated in the magnetic field of about 10 reported by Webster and Ziebeck [1], Buschow et al. [16], kOe. The magnetization increases with increas and Buschow and van Engen [17]. The concentration spontaneous magnetization  $\sigma_s$  at 5 K was determined by dependence of the lattice parameter *a* at room temperature the linear extrapolation to  $H/\sigma = 0$  of the  $\sigma^2$  ver for Fe<sub>2+x</sub>V<sub>1-x</sub>Al ( $0 \le x \le 0.4$ ) is shown in Fig. 2. The curves (Arrott plot) at high fields. The concentration lattice parameter increases with increasing Fe concen-<br>tependence of  $\sigma_s$  at 5 K is shown in Fig. 4 with the data tration. The V atom in Fe, VA1 has eight Fe nearest reported by Endo et al. [3]. The spontaneous magnetiza neighbors and the Fe atom has four V and four Al nearest of  $Fe_{2+x}V_{1-x}Al$  increases linearly with increasing *x*. The neighbors. For Fe<sub>2+x</sub>V<sub>1-x</sub>Al, the selective substitution of magnetic moment at 5 K of Fe<sub>2.05</sub> V<sub>0.95</sub>Al is found to be Fe atoms for the V site in Fe<sub>2</sub>VAl was supported by X-ray 0.26  $\mu_R$  per formula unit. Assuming th Fe atoms for the V site in Fe<sub>2</sub>VAI was supported by X-ray 0.26  $\mu_B$  per formula unit. Assuming that excess Fe atoms analysis [14] and band calculation [8]. We have confirmed at the V sites in Fe<sub>2.05</sub>V<sub>0.95</sub> AI form 9Fe the atomic ordering for all samples from the appearance of carry magnetic moments, the magnetic moment per cluster



fields up to 50 kOe. As seen in Fig. 3, the magnetization  $\sigma$ kOe. The magnetization increases with increasing  $x$ . The reported by Endo et al. [3]. The spontaneous magnetization at the V sites in Fe<sub>2.05</sub> $V_{0.95}$  Al form 9Fe atom clusters and the superlattice lines in X-ray diffraction spectra. is found to be 5.2  $\mu_B$ . Recently, Bansil et al. [11] have Magnetization data were taken using a commercial carried out the band calculation for  $Fe_{2+x}V_{1-x}Al$  ( $x=0.25$ , superconducting quantum interference device (SQUID) 0.50 and 0.75), together with the end compounds Fe<sub>3</sub>Al magnetometer. The DC electrical resistivity ( $\rho$ ) was and Fe<sub>2</sub>VAl, and the limiting cases of a single V impurit and Fe<sub>2</sub>VAl, and the limiting cases of a single V impurity measured by a four-probe technique using mechanical in  $Fe_3$ Al and a single Fe impurity at the V sites in  $Fe_2$ VAI positional point contacts, where tungsten wires (0.3 mm Ø) using the methodology based on the Korringer–Ko using the methodology based on the Korringer–Kohn– are used as spring for pressure contact and platinum is Rostoker formalism and the coherent-potential approxisoldered on the end of the tungsten wires to ensure the mation [11]. In the  $x=0$  limit (the dilute Fe impurity



centrations. data from Ref. [2], respectively. The solid line in the figure is a guide for

limit), the magnetic moment on Fe atom at the V sites was found to be 3.2  $\mu_B$ . Therefore, 8Fe atoms surrounding the<br>center atom in a cluster may be expected to carry a small<br>magnetic moment for Fe<sub>2.05</sub>V<sub>0.95</sub>Al.<br>Fig. 5 shows the concentration dependence of  $T_c$  for<br>Fe<sub>2+x</sub>V<sub>1</sub>

et al. [2]. In this figure our results, represented by the



results and the data from Ref. [3], respectively.



Fig. 5. Concentration dependence of the Curie temperature  $T_c$  for Fig. 3. Magnetization curves at 5 K of Fe<sub>2+x</sub>V<sub>1-x</sub>Al at various con-Fe<sub>2+x</sub>V<sub>1-x</sub>Al. The closed and open circles are the present results and the the eye.



Fig. 4. Concentration dependence of spontaneous magnetization  $\sigma_s$  at 5 K Fig. 6. Temperature dependence of the electrical resistivity  $\rho$  for for Fe<sub>2+x</sub>V<sub>1-x</sub>Al. The closed circles and closed triangles are the present  $Fe_{2+x}V_{1-x}Al$  for various concentrations. The arrows indicate the Curie temperature  $T_c$ .

cal resistivity  $\rho$  in Fe<sub>2+x</sub>V<sub>1-x</sub>Al with various concentrations. As seen in the figure,  $\rho$  increases with increasing V concentration. The samples with  $x \le 0.10$  exhibit a semiconductor-like behavior in the temperature range investigated. We can estimate the energy gap  $E_G$  of Fe<sub>2</sub>VAl to be 0.0847 eV from the slope of ln  $\rho$  versus  $1/T$  curve between 300 and 550 K. This value is somewhat smaller than that  $(E_G \sim 0.1 \text{ eV})$  reported by Nishino et al. [2]. Endo et al. [3] reported the value of  $E_G = 0.07$  eV for Fe<sub>2</sub>VAl. For the data of  $Fe_{2+x}V_{1-x}Al$  ( $x=0$ , 0.02, 0.05, 0.08 and 0.10), the ln  $\rho$  versus  $1/T$  plots are linear in the temperature range from  $\sim$ 350 to  $\sim$ 570 K. The concentration dependence of the energy gap deduced from  $\ln \rho$  versus  $1/T$  is shown in Fig. 7. The energy gap decreases linearly with increasing the concentration *x*. For  $x=0.18$ , 0.25 and 0.40,  $\rho$  makes a broad maximum at  $T_c$  or near  $T_c$  as shown in Fig. 6. The arrows in Fig. 6 indicate  $T_c$ determined from the thermomagnetic measurements.

Fig. 8 shows the temperature dependence of the thermoelectric power *S* for  $Fe<sub>2.18</sub>V<sub>0.82</sub>Al.$  As shown in the figure, *S* decreases with decreasing temperature and makes a minimum at 400 K, and then increases. The small Fig. 8. Temperature dependence of the thermoelectric power *S* for negative din is observed at  $T_c$ . This behavior of the  $S$  Fe<sub>2.18</sub>V<sub>0.82</sub>Al. negative dip is observed at  $T_c$ . This behavior of the *S* versus *T* curve suggests that the Fermi surface is very sensitive to temperature. The results was done by de Gennes and Friedel [19]. By using

Fe<sub>2.18</sub>V<sub>0.82</sub>Al is shown in Fig. 9.  $\Delta \rho / \rho(0)$  for Fe<sub>2.18</sub>V<sub>0.82</sub>Al action of the form  $V = G \sum \delta(r - R_j)S_j \cdot s$  between the spin<br>is always negative and reaches 20.5% at 220 kOe, while it of the atom  $S_j$  at position  $R_j$  is not yet saturated. The temperature dependence of  $\Delta \rho$  electron s, they showed that at temperatures well above  $I_c$ <br> $\rho(0)$  with various fields is shown in Fig. 10. The absolute the magnetic part of the resistivity  $\r$ values of  $\Delta \rho / \rho(0)$  are maximum near  $T_c$  as indicated by independent and at  $T_c$  decreases with decreasing temperature. The temperature dependence of  $\rho$  at  $T < T_c$  is given

A theoretical treatment applicable to our experimental



The high-field transverse magnetoresistance  $\Delta \rho / \rho(0)$  for the Born approximation and assuming an exchange inter-



Fig. 7. Energy gap  $E_G$  versus concentration *x* curve for  $Fe_{2+x}V_{1-x}Al$ . Fe<sub>2.18</sub>V<sub>0.82</sub>Al at various temperatures.



Fig. 9. Field dependence of the transverse magnetoresistance  $\Delta \rho / \rho(0)$  for



Fig. 10. Temperature dependence of the transverse magnetoresistance [8] G.Y. Guo, G.A. Botton, Y. Nishino, J. Phys. Condens. Matter 10  $|\Delta \rho / \rho(0)|$  for Fe<sub>2.18</sub> V<sub>0.82</sub> Al at various fields. The arrow indicates the Curie (1998) L119. temperature  $T_c$  determined from the thermomagnetic measurement. [9] D.J. Singh, I.I. Mazin, Phys. Rev. B 57 (1998) 14352.

by  $\rho_o^{\text{mag}}\{1 - \langle S \rangle^2 / S(S + 1)\}$ , where S is the spin of the [11] A. Bansil, S. Kaprzyk, P.E. Mijnarends, J. Tobala, Phys. Rev. B 60 magnetic atom. Furthermore, they showed that when there exists the short range order in  $T_c$ ,  $\rho$  versus *T* curve makes a cusp near  $T_c$ . This idea was [14] E. Popiel, M. Tuszynski, W. Zarek, T. Rendecki, J. Less-Common derived by taking account of the scattering of the conduc-<br>
tion electrons due to the spin fluctuations It should be [15] J.B. Nelson, D.P. Rily, Proc. Phys. Soc. Lond. 57 (1945) 160. tion electrons due to the spin fluctuations. It should be<br>noted that  $\rho$  versus T curves of Fe<sub>2+x</sub>V<sub>1-x</sub>Al (x=0.18,<br>0.25 and 0.40) obtained in this study show small maximum<br> $\frac{[15] \text{ J.B. Nelson, D.P. Rily, Proc. Phys. Soc. Lond. 57 (1945) 160.}}{$ though the sharp cusp as proposed by de Gennes and 90. Friedel was not observed in  $\rho$  versus *T* curves. The [18] E. Nagy, J. Toth, J. Phys. Chem. Solids 24 (1963) 1043.<br>negative giant magnetoresistance near *T* of Ee V A1 [19] P.G. de Gennes, J. Friedel, J. Phys. Chem. Sol negative giant magnetoresistance near  $T_c$  of Fe<sub>2.18</sub>V<sub>0.82</sub>Al may be due to the suppression of the large spin fluctuations by external magnetic fields.

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